

# Molecular Arrangement in C<sub>60</sub> and C<sub>70</sub> Films on Graphite and Their Nanotribological Behavior

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Received December 7, 2000

## ABSTRACT

The nanotribological behavior of C<sub>60</sub> and C<sub>70</sub> films on graphite has been studied. C<sub>60</sub> molecules on graphite begin to grow in a monolayer form, whereas C<sub>70</sub> molecules begin to grow in a bilayer form. The shear stress between a C<sub>60</sub> monolayer and graphite was estimated to be about 0.2 GPa. For C<sub>60</sub>(111) films, a change of tip motion, from one-dimensional stick-slip to two-dimensional zigzag stick-slip with decrease of a loading force, appears.

The mechanical properties of fullerenes have attracted much attention in the field of materials science. Particularly, C<sub>60</sub> and C<sub>70</sub> solids have been predicted to be good lubricant materials because of their nearly spherical shape and low surface strength.<sup>1–3</sup> Our previous letter<sup>3</sup> reported on the nanotribological behavior of the C<sub>60</sub> islands on KCl(001) using frictional force microscopy. Along the different scanning directions of  $\langle 1\bar{1}0 \rangle$  and  $\langle 11\bar{2} \rangle$  of a C<sub>60</sub>(111) surface, a tip has performed one-dimensional stick-slip and zigzag stick-slip motions on the order of a load of nN, respectively, although at a larger loading force, it destroyed the C<sub>60</sub>(111) surface. Also, water adsorption on the C<sub>60</sub> films gave a lower frictional force, indicating that C<sub>60</sub> molecules rotate or translate at the (111) surface. Thus, the C<sub>60</sub> films were expected to exhibit various behaviors depending on the loading force, scanning direction, and relative humidity.

In this letter, we report the growth mode of C<sub>60</sub> and C<sub>70</sub> films on graphite using atomic force microscopy and frictional force microscopy, and we focus on the nanotribological properties of C<sub>60</sub> and C<sub>70</sub> films.

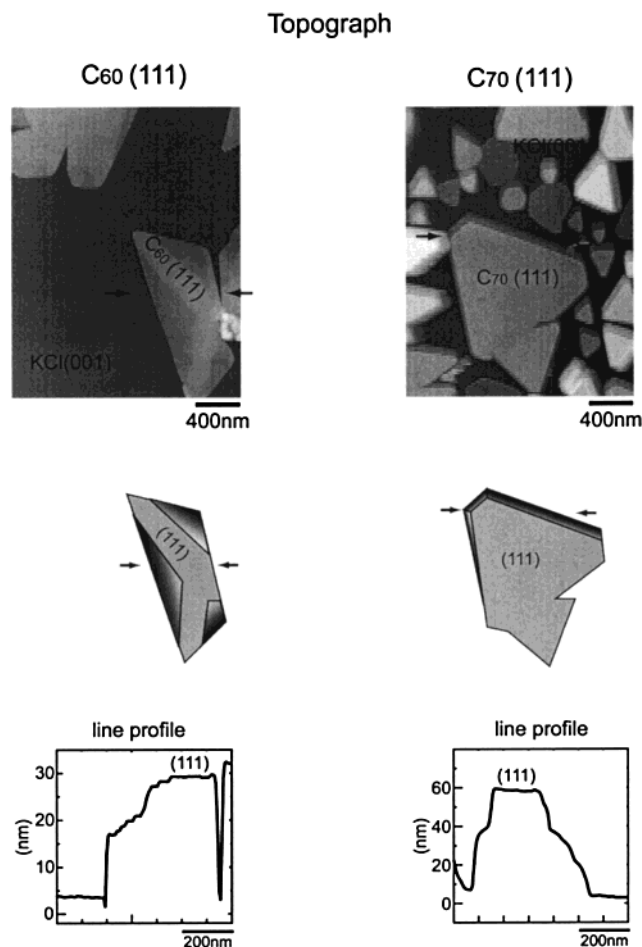
The C<sub>60</sub> and C<sub>70</sub> films on highly oriented pyrolytic graphite (HOPG) and KCl(001) were prepared by evaporation from a BN crucible. The temperatures of their substrates during evaporation were kept at range of 150 °C to 200 °C. Normal and lateral forces were measured simultaneously under argon atmosphere at room temperature using a commercially available instrument (Seiko Instruments Inc., SPI-3700). The scan speed was 0.13 m/s. A rectangular silicon cantilever

with a normal spring constant of 0.05 N/m was used. Zero normal force is defined as the position at which the cantilever is not bent.

Figure 1 shows the atomic force microscope (AFM: topograph) images of C<sub>60</sub> and C<sub>70</sub> islands on a KCl(001) surface. Their islands consist of scores of molecular layers. The (111) facets of the C<sub>60</sub> and C<sub>70</sub> islands appear, which form parallel to the KCl(001) substrate. Such island growth occurs in cases where an intermolecular interaction is stronger than a molecule–substrate interaction, as reported in previous papers.<sup>4–6</sup> Thus, it is concluded that C<sub>60</sub>–C<sub>60</sub> and C<sub>70</sub>–C<sub>70</sub> interactions are stronger than C<sub>60</sub>–KCl(001) and C<sub>70</sub>–KCl(001) interactions, respectively. Figure 2 shows the AFM (topograph) images of C<sub>60</sub> and C<sub>70</sub> films on a graphite surface at the initial stage of growth. The height of the C<sub>60</sub> films on the middle of Figure 2 is approximately 1 nm, exhibiting a monolayer. As shown at the bottom of Figure 2, the high-resolution frictional force microscope (FFM) image of the C<sub>60</sub> monolayer has a periodicity of 0.9 nm, which shows a zigzag motion along the  $\langle 11\bar{2} \rangle$  scanning direction of the C<sub>60</sub> monolayer.<sup>3</sup> Thus, the existence of the monolayer exhibits epitaxial growth on graphite. As shown by Gravié et al.,<sup>7</sup> the growth of this monolayer indicates that C<sub>60</sub> molecules grow on graphite such that the hexagonal face of a C<sub>60</sub> molecule pairs with the hexagonal face of the second graphite layer, so as to continue the natural stacking (AB stacking) of the graphite. The height of the C<sub>70</sub> films in the middle of Figure 2 is about 2 nm, exhibiting a bilayer. Here, it should be noted that there does not exist any C<sub>70</sub> monolayer on graphite in our experiments. Because the hexagonal face in a C<sub>70</sub> molecule does not situate on the top of its ellipsoid, as shown on the middle of Figure 2, it may not always be

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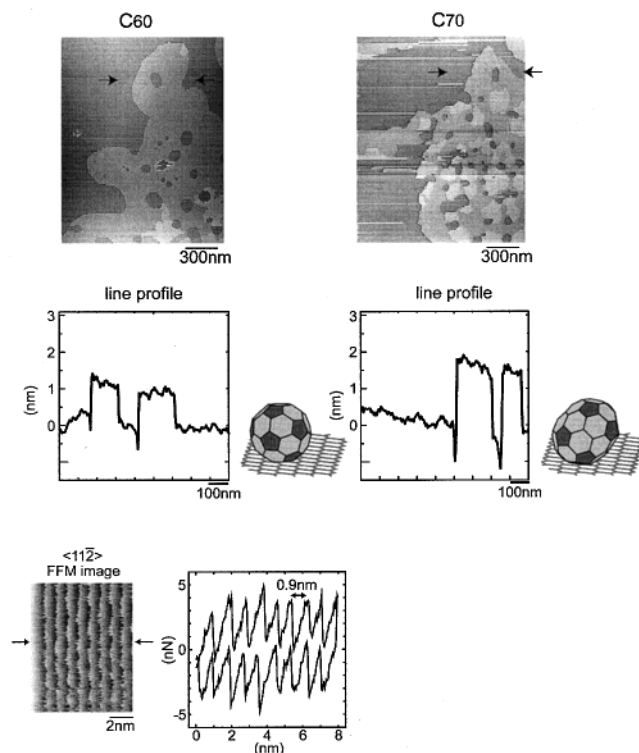
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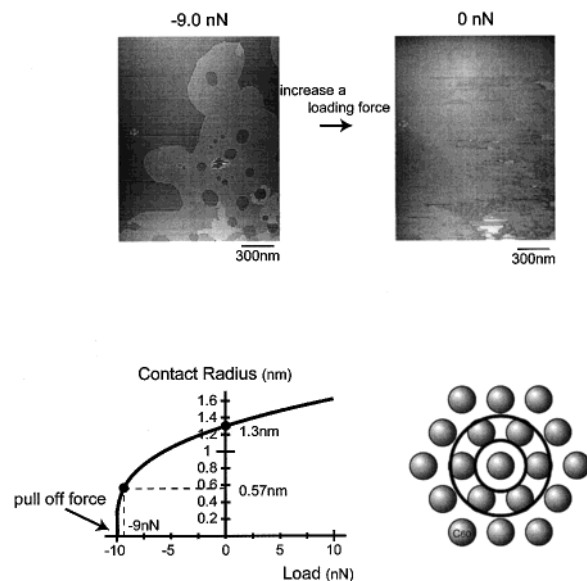
**Figure 1.** Atomic force microscope (AFM: topograph) images of  $C_{60}$  and  $C_{70}$  islands on a KCl(001) surface. The (111) facets of the  $C_{60}$  and  $C_{70}$  islands appear, which form parallel to a KCl(001) substrate. At the bottom, line profiles depicted by the arrows are illustrated.

stable to pair with the hexagon face of the graphite in a monolayer manner.

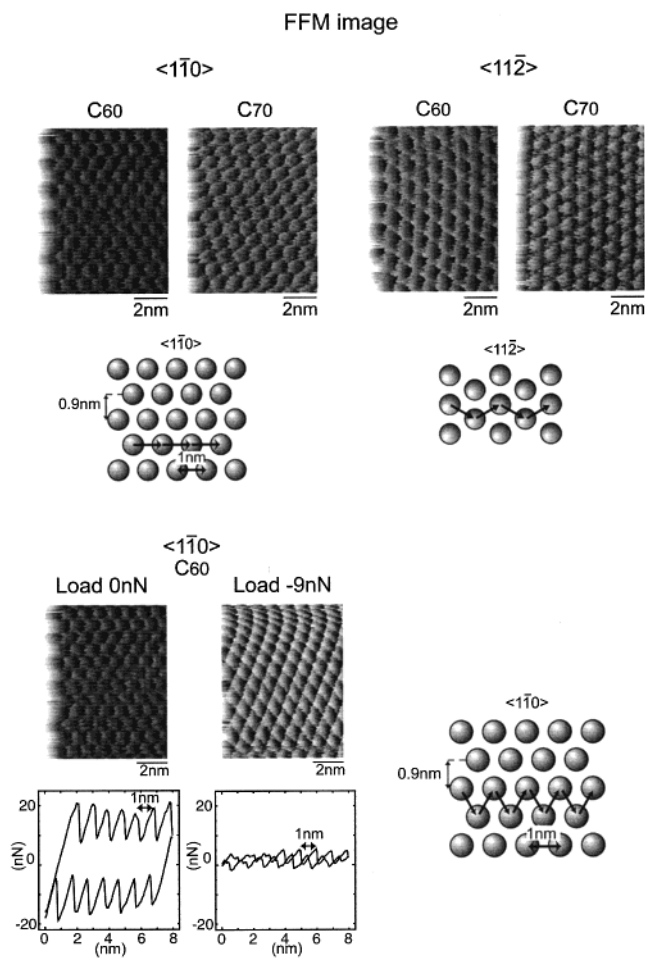
Figure 3 shows how the AFM image from the  $C_{60}$  monolayer of Figure 2 changes by increasing a loading force. One notes that the  $C_{60}$  monolayer on graphite is swept up by increasing up to a load of 0 nN. Using the Hertzian continuum theory,<sup>8,9</sup> the contact radius versus the tip load estimated using a pull-off force of  $-10$  nN and a tip radius of  $15$  nm is shown on the bottom of Figure 3. At a loading force of  $0$  nN, the contact radius between the tip and the  $C_{60}$  monolayer surface is estimated to be about  $1.3$  nm, exhibiting that the contact area between the tip and the  $C_{60}$  monolayer is seven molecular dimensions. Then, the mean lateral force was about  $3$  nN. Thus, assuming that seven molecules are moved by a lateral force of  $3$  nN with the tip, the mean shear force between the single  $C_{60}$  molecule and the graphite is estimated to be about  $0.4$  nN, such that the shear stress between the  $C_{60}$  monolayer and the graphite is estimated to be about  $0.2$  GPa. For similar experiments with  $C_{70}$  films on graphite, the  $C_{70}$  films were easily swept up at loading forces smaller than those of the  $C_{60}$  films. Thus, this exhibits that the interaction between the  $C_{70}$ -graphite interfaces is weaker than that between the  $C_{60}$ -graphite interfaces.



**Figure 2.** AFM (topograph) images of  $C_{60}$  and  $C_{70}$  films on a graphite surface at the initial stage of growth. In the middle, line profiles depicted by the arrows are illustrated. The height of  $C_{60}$  films is about  $1$  nm, exhibiting a monolayer. At the bottom of the figure, the high-resolution frictional force microscope (FFM) image of the  $C_{60}$  monolayer has a periodicity of  $0.9$  nm, which shows a zigzag motion along the  $\langle 11\bar{2} \rangle$  scanning direction of the  $C_{60}$  monolayer. The height of  $C_{70}$  films is about  $2$  nm, exhibiting a bilayer.



**Figure 3.** Change of the AFM (topograph) image from the  $C_{60}$  monolayer of Figure 2 with an increase in the loading force. One notes that the  $C_{60}$  monolayer on graphite is swept up by increasing up to a load of  $0$  nN. Using the Hertzian continuum theory, the contact radius versus the loading force was estimated and is shown at the bottom, where the pull-off force is  $-10$  nN. Contact radii at loading forces of  $0$  nN and  $-9$  nN are  $1.3$  nm (solid circle) and  $0.57$  nm (dotted circle), respectively.



**Figure 4.** High-resolution FFM images from (111) surfaces of  $C_{60}$  and  $C_{70}$  films. For the  $\langle 1\bar{1}0 \rangle$  and  $\langle 1\bar{1}2 \rangle$  scanning directions, the samples exhibit one-dimensional stick-slip and two-dimensional zigzag motions, respectively. In the case of the  $\langle 1\bar{1}0 \rangle$  scanning direction of the  $C_{60}$  (111) films on the bottom of the figure, it should be interesting to note that the image at a loading force of  $-9$  nN (pull-off force,  $-10$  nN) changes from one-dimensional stick-slip to clear two-dimensional motions, exhibiting a single molecular contact with the tip, which gives a mean frictional force of  $1$  nN.

This result is consistent with the assumption that is stated above, that the  $C_{60}$ -graphite interfaces are composed of the

natural stacking (AB stacking), whereas the  $C_{70}$ -graphite interfaces are not so composed.

Figure 4 shows the high-resolution FFM images from (111) surfaces of  $C_{60}$  films and  $C_{70}$  films. For the  $\langle 1\bar{1}0 \rangle$  and  $\langle 1\bar{1}2 \rangle$  scanning directions, the films exhibit one-dimensional stick-slip and two-dimensional zigzag motions, respectively, which is discussed in detail in the previous paper.<sup>3</sup> In the case of the  $\langle 1\bar{1}0 \rangle$  scanning direction of the  $C_{60}$  (111) films, it is interesting to note that the image at a loading force of  $-9$  nN (pull-off force,  $-10$  nN) changes from one-dimensional stick-slip to clear two-dimensional motions, exhibiting a single molecular contact with the probe tip, which gives a mean frictional force of  $1$  nN.<sup>10</sup> In other words, this is a phase transition of stick-slip motion due to a change from a multi-molecular contact to a single molecular contact. Thus, the shear force between the tip and the single  $C_{60}$  molecule is estimated to be  $1$  nN. It should be noted that the shear force ( $1$  nN) between the tip and the  $C_{60}$  molecule is larger than that ( $0.4$  nN) between the  $C_{60}$  molecule and the graphite. This is consistent with the experimental result that a tip scanning at the  $C_{60}$  monolayer or bilayer is unstable, whereas a tip scanning at several  $C_{60}$  layers is stable.

**Acknowledgment.** We thank T. Sahashi and S. Kamiya for their assistance with the experiments.

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NL0055308